



Chair of Condensed Matter Physics
Institute of Theoretical Physics
Faculty of Physics, University of Warsaw

Summer Semester 2013

Lecture


Modeling of Nanostructures and Materials

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Modeling of Nanostructures and Materials SS 2013

Jacek A. Majewski

- Lecture for  With the basis knowledge of **Quantum Mechanics**
- **Lecture --- 45 h**
- **Practical exercises – 45 h**
- Exam, elements of the note:
 - Test from lecture material,
 - Work during excercises
 - Project
- **Aim of the lecture: make familiar with modern modeling tools (ability to perform calculations with standard tools)**

Modeling of Nanostructures and Materials SS 2013

Scope of the lecture

Ab initio Methods:

- Density Functional Theory,
- Local Density Approximation (LDA),
- Hartree-Fock method,
- Kohn-Sham Method,
- Concept of Pseudopotential,
- Survey of available numerical codes.

Modeling of Nanostructures and Materials SZ 2013

Scope of the Lecture (cnt.)

Semiempirical Methods for electronic structure calculations:

- Tight-Binding Method
- Pseudopotential Method

Principles of Molecular Dynamics:

- *Ab initio* molecular dynamics (Car-Parrinello method)
- Empirical methods and coarse-graining
- Valence-force field models

Monte Carlo Methods:

- Stochastic and Markov processes, ergodicity,
- Algorithms for Monte Carlo simulations



Modeling of Nanostructures and Materials

Jacek A. Majewski

Lecture 1 – 21 February 2013

Introduction to

Computational Science,

(Computer Simulations, Computer Modeling),

Computational Materials Science

Modeling of Nanostructures

Why should we bother with it?

Computational Science

- a relatively new discipline
- involves using computers to study *scientific problems*
- complements the areas of theory and experimentation in traditional scientific investigation
- seeks to gain understanding of science principally through the use and analysis of mathematical models on (high performance) computers
- emerged as a powerful and indispensable method of analyzing a variety of problems in research, product and process development, and manufacturing.

➔ **Computational Simulations**
Changing the way we do Science ?

Computational Simulations

- *Computer simulations* provide both qualitative and quantitative insights into many phenomena that are too complex to be dealt with by analytical methods or too expensive or dangerous to study by experiments.
- Many experiments and investigations that have traditionally been performed in a laboratory, a wind tunnel, or the field are being augmented or replaced by computer simulations.
- Some studies, such as nuclear repository integrity and global climate change, involve time scales that preclude the use of realistic physical experiments.

Computational Simulations

The availability of

- high performance computers,
- graphic workstations,
- and high speed networks,

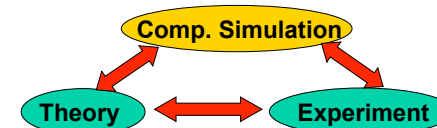
coupled with major advances in algorithms and software, has brought about a revolution in the way scientific and engineering investigations are carried out.

Computational Science vs. Computer Science

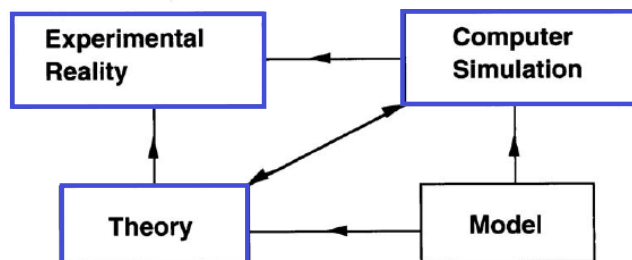
- These two things should not be confused !
- **Computational science** focuses on a scientific or engineering problem and draws from computer science and mathematics to gain an improved understanding of the problem.
- **Computer science** focuses on the computer itself.
- **However,**
Even though the areas are quite distinct, many of the topics typically considered to be in the domain of computer science are of much value in computational science.

Computational Sciences

- Computational Physics
- Computational Astrophysics and Cosmology
- Computational Geophysics
- Computational Chemistry
- Computational Biology
- Computational Engineering
- Computational Materials Science
- Computational Nanoscience (Modeling of Nanostructures)



Relationship between modeling, theory and experiment



Dr P. D. Bristowe, University of Cambridge,
Department of Materials Science and Metallurgy

The computer - the hallmark of the information age

- Computational power has been compounding at an exponential rate for nearly 40 years,
 - an achievement made possible by advances in silicon processing and fabrication at ever smaller length scales.
- Breakthroughs in control and fabrication of magnetic media have driven disk capacity to likewise grow exponentially.

Importance of Materials for Society

The computer is possibly the most visible example of a high tech product that depends critically upon advanced materials

There are many other similar examples that also profoundly affect our lives,

- new lightweight alloys,
- polymers,
- composites, etc.

➔ *Materials Science*

Computational Materials Science

- From the Bronze Age to the silicon-driven Information Age, civilization has defined itself—and advanced itself—by mastering new materials.
- The ability to identify, improve and implement materials -- whether stone, iron, paper, steel or silicon -- has profoundly shaped human societies.
- Today, thanks to increasingly powerful computers, the materials science community finds itself on the verge of another revolution.

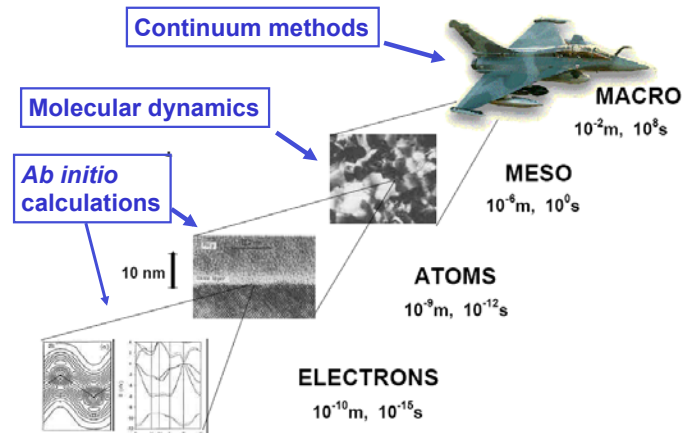
➔ **extensive computational modeling will complement and sometimes even replace traditional methods of trial-and-error experimentation.**

➔ **better understanding and design of new materials**

Computational Materials Science - Scales

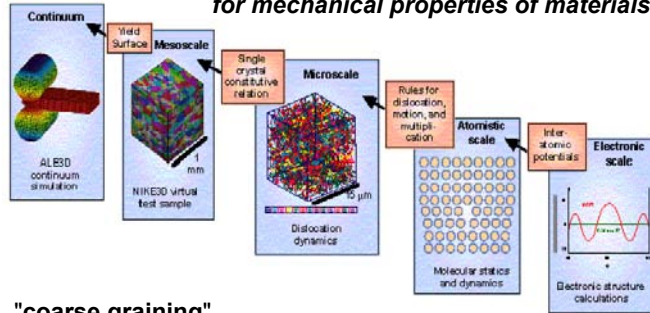
- Scientists have a handle on the smallest length scale (relevant to materials), which cannot be seen with a microscope, and the largest length scale, which can be seen with the naked eye.
- In between is an intermediate length scale where there are particularly exciting new frontiers.
- The primary scientific challenge is to uncover the elusive connections in the hierarchy of time and length scales and to unravel the complexity of interactions that govern the properties and performance of materials.

Connection of atomistic and macroscopic scales



Computational Materials Science – Multi-scale Simulations

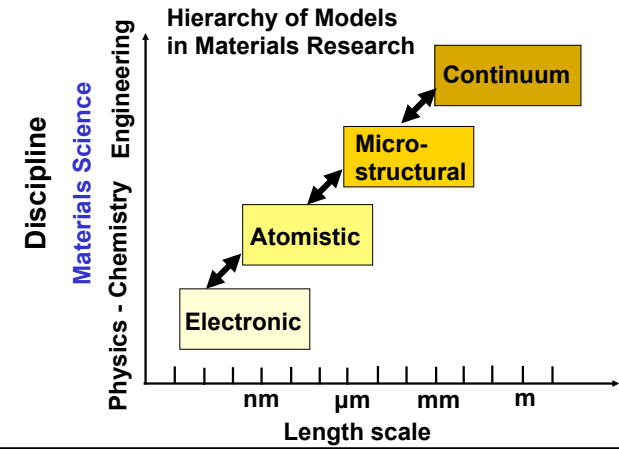
A schematic of the multi-scale challenge for mechanical properties of materials



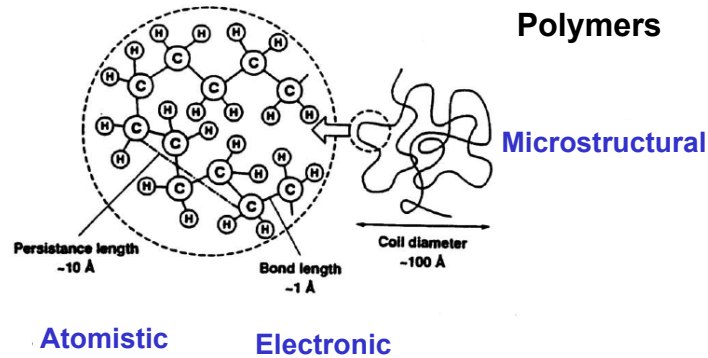
"coarse graining"

the forces among the aggregated pieces can be accurately described by a new model with "renormalized" interactions.

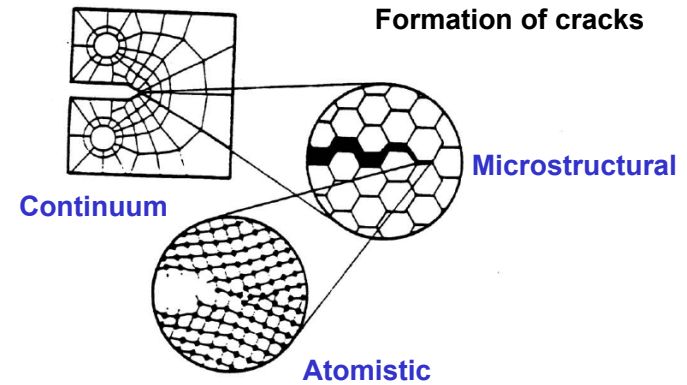
Types of Physical Models Based on Length Scale



Examples of Models Based on Length Scale (1)



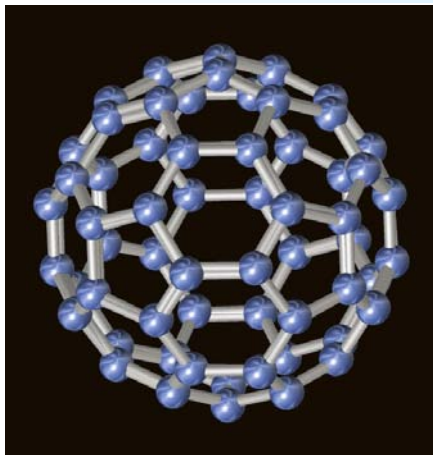
Examples of Models Based on Length Scale (2)



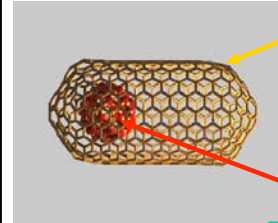
Examples of Computational Simulations

- Future nano-electronics
- Chemical reactions
- Computational biology

Fullerene (buckyball)



Simulation of a nanotube-based memory element

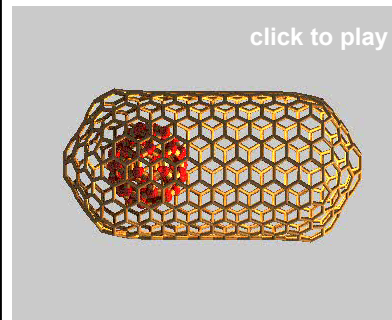


The outer capsule - short segment of (10,10) carbon nanotube with diameter 1.4 nm terminated at both ends by perfect fullerene caps.

The capsule is large enough to hold a C_{60} buckyball molecule inside.

- The buckyball carries a net charge if it contains an alkali atom in the hollow cage.
- The C_{60} molecule can be shifted from one end to the other by applying an *electric field* between the ends of the capsule

Simulation of a nanotube-based memory element



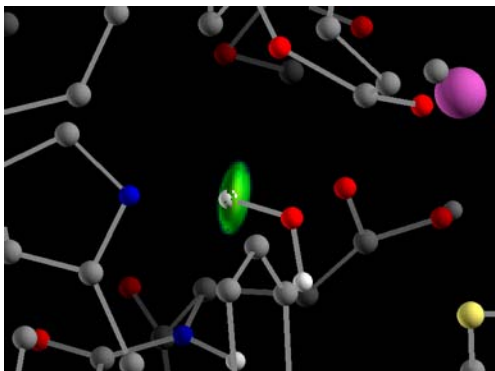
The **two energy minima** of this system, with the buckyball bonded to either end of the capsule, can be associated with **bit 0** and **bit 1**.

The simulation has been performed by Young-Kyun Kwon

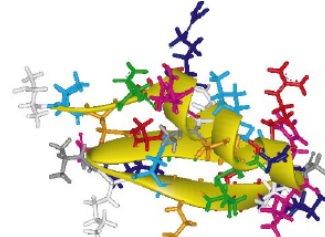
This structure has been patented as a non-volatile memory element and awarded U.S. Patent No. 6,473,351

Ab initio MD Simulations of Chemical Reaction

Dissociation of water molecule

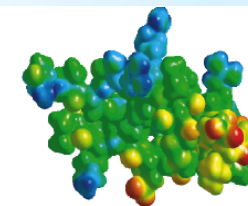


DFT as a new tool for computational biology



Structure of the 1PNH protein, a scorpion toxin ($\text{PO}_5\text{-NH}_2$)

- It consists of 31 amino acid Residues (ca. 500 atoms)
- Different colors are used to represent different amino acids



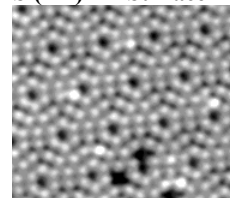
- ### Electrostatic potential for electrons
- Zero value – green
 - Repulsive regions – red and yellow
 - Attractive regions - blue

Role of the protein charge state on the geometry – change of the charge state of the protein does not destabilize the local energy minima

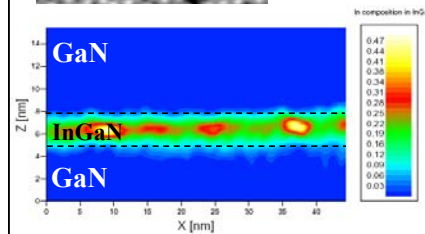
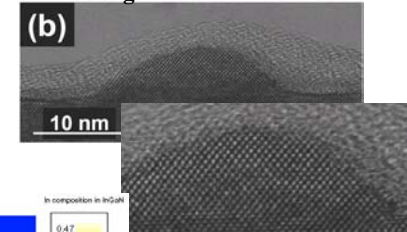
Modeling Nanostructures

Examples of Nanostructures

Si(111)7×7 Surface



TEM image of a InAs/GaAs dot

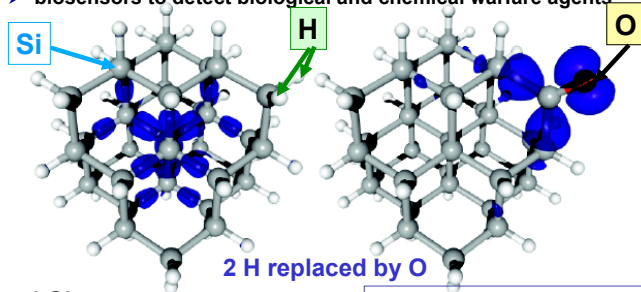


HRTEM image: segregation of Indium in GaN/InGaN Quantum Well

DFT for silicon nanostructures

Silicon nanoparticles (clusters, dots)

- optoelectronic materials on silicon basis
- biosensors to detect biological and chemical warfare agents



- 71 Si atoms
- 'passivated' by hydrogens
- Electrons are in the center of the dot

Dramatic change of the optical properties (wavelength) of the silicon nanostructure

G. Gali & F. Gygi, Lawrence Livermore National Laboratory

What about realistic nanostructures ?

Inorganics

3D (bulks) : 1-10 atoms in the unit cell

2D (quantum wells): 10-100 atoms in the unit cell

1D (quantum wires): 1 K-10 K atoms in the unit cell

0D (quantum dots): 100K-1000 K atoms in the unit cell

Organics

Nanotubes, DNA: 100-1000 atoms (or more)

Atomistic methods for modeling of nanostructures

- **Ab initio methods** (up to few hundred atoms)
- **Semiempirical methods** (up to 1M atoms)
 - Empirical Pseudopotential
 - **Tight-Binding Methods**

Tight-Binding methods

Tight-Binding Formalism

$$\varphi_n(\vec{r}) = \sum_{\alpha,i} c_n^{\alpha i} \chi_{\alpha i}(\vec{r})$$

index of orbital

index of atom

$\{ \chi_{\alpha i} \}$ orthogonal set of functions
 NOT ATOMIC ORBITALS!

$$\varepsilon_n = \langle \varphi_n | H | \varphi_n \rangle = \sum_{i\alpha} \sum_{j\beta} (c_n^{\alpha i})^* c_n^{\beta j} \langle i\alpha | H | j\beta \rangle$$

Tight-binding Hamiltonian

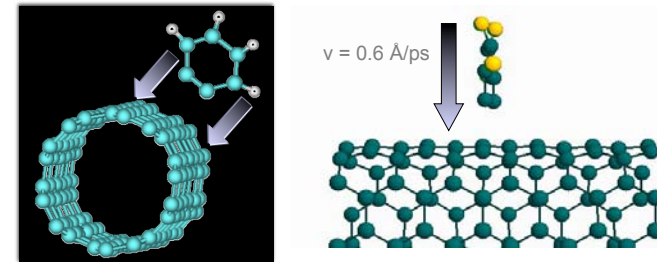
$$H = \sum_{\alpha i} \varepsilon_{i\alpha} | i\alpha \rangle \langle i\alpha | + \sum_{\alpha i, \beta j} t_{i\alpha, j\beta} | i\alpha \rangle \langle j\beta |$$

on-site

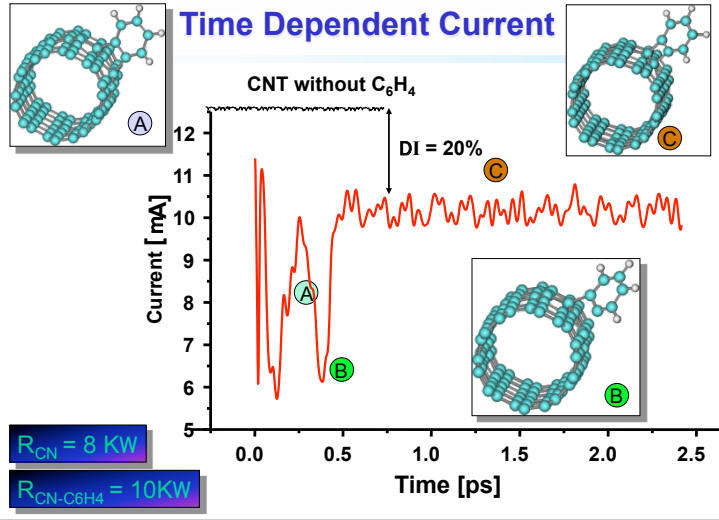
hopping TB-parameters

Green's Function + Molecular dynamics Carbon Nanotubes

- **Molecular Dynamics** simulations of a reactive collision of a biased nanotube ($V=100\text{mV}$) and benzene
- **Current** flowing in the nanotube calculated at each MD step

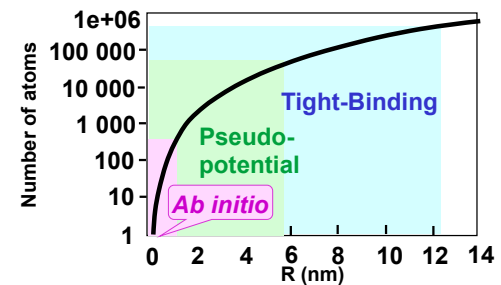


Time Dependent Current



Length scale for *ab initio* and tight-binding methods

- Microscopic approaches can be applied to calculate properties of realistic nanostructures



Number of atoms in a spherical Si nanocrystal as a function of its radius R. Current limits of the main techniques for calculating electronic structure. Nanostructures commonly studied experimentally lie in the size range 2-15 nm.

Computational Materials Science – The Era of Applied Quantum Mechanics

The properties of new and artificially structured materials can be predicted and explained

- *entirely by computations,*
- *using atomic numbers as the only input.*

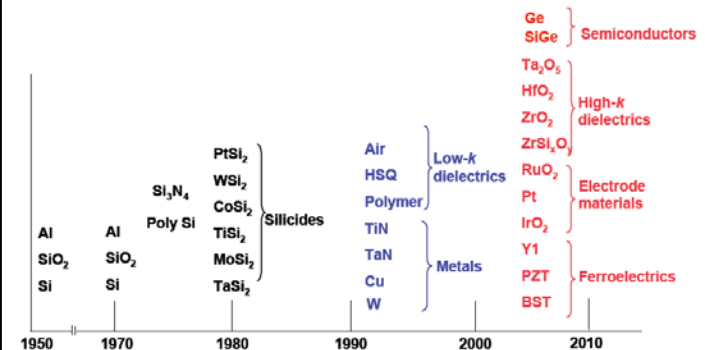
Computational Materials Science: A Scientific Revolution about to Materialize

- The materials science community is on the verge of a paradigm shift in the way it does science with the promise of building a sturdy bridge across the "valley of death" between basic science and technological impact.
- A useful construct for thinking about this potential paradigm is "**Pasteur's Quadrant.**"

D. E. Stokes, "Pasteur's Quadrant, Basic Science and Technological Innovation,,
The Brookings Institution, Washington D.C., 1997

Recent technological developments
cause increasing demands
for materials with specific properties

New Materials for Si- Technology

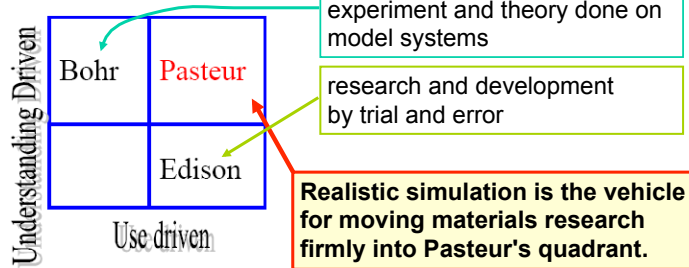


- **Moore's Law increasingly relies on material innovations !**

Computational Materials Science: A Scientific Revolution about to Materialize

- Due to the complexity of materials systems, progress has necessarily proceeded either within the Bohr quadrant or Edison's quadrant

Pasteur's Quadrant



Recent technological developments cause increasing demands for materials with specific properties

- Experiments are without doubt the most important approach in studying materials
- However, Kohn and co-workers opened a new avenue to *study properties of materials from first-principles with the formulation of density functional theory (DFT)*.

P. Hohenberg, W. Kohn, *Phys. Rev.* **136**, B864 (1964).

Density Functional Theory – the key to Computational Materials Science & Multiscale Modeling of Nanostructures

The Basics

Usage of Density Functional Theory (DFT)

- Using DFT many practical problems of materials science have been solved successfully.
- DFT is now employed not only by physicists, but also by chemists, geophysicists, biophysicists, metallurgist, and in other scientific fields.
- The computational implementations of DFT together with modern solid state theory allow it to obtain reliable results for *thermodynamic, mechanical, electrical and magnetic properties* of
 - metals,
 - semiconductors, or
 - insulators*without any adjustable parameters* fitted to the experiment.

DFT – Applied to real materials

- There are numerous applications of DFT in chemistry and physics (> 10 000 papers a year)
- The use of DFT based methods is still very new in the field of engineering
- The simulation of cracks in materials belongs to the most challenging problems in materials science.
- While the crack itself is a macroscopic property, the physical processes at the crack tip itself involve **the breaking of bonds governed by quantum mechanics**.

DFT – Nobel Prize in Chemistry, 1998

The big impact of DFT has been clearly high lightened by awarding the Nobel Prize in Chemistry in 1998 for the **development** and **application** of **DFT**.

Walter Kohn



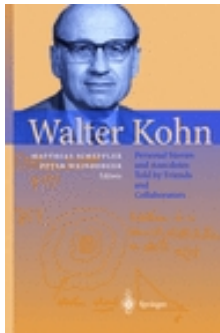
Born in 1923

John A. Pople



1925 - 2004

80th birthday of Walter Kohn



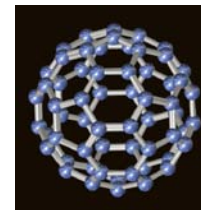
“Walter Kohn –
Personal Stories and Anecdotes
Told by Friends and Collaborators”

eds. Matthias Scheffler &
Peter Weinberger

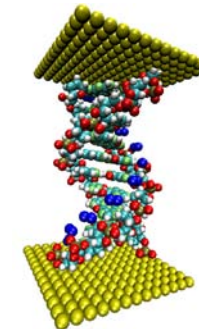
Springer Verlag

Fundamental problem in materials science

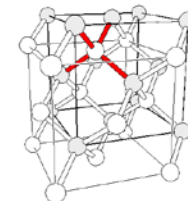
A fundamental problem in materials science is
the prediction of condensed matter's electronic structure



C₆₀ - molecule



DNA - molecule



Crystal - diamond

Materials Science: Examples of Schrödinger Equation?

- Materials are composed of nuclei $\{Z_\alpha, M_\alpha, \vec{R}_\alpha\}$ and electrons $\{\vec{r}_i\}$
→ the interactions are known

$$H = -\sum_{\alpha} \frac{\hbar^2 \nabla_{\alpha}^2}{2M_{\alpha}} - \sum_i \frac{\hbar^2 \nabla_i^2}{2m} + \frac{1}{2} \sum_{\alpha, \beta} \frac{Z_{\alpha} Z_{\beta} e^2}{|\vec{R}_{\alpha} - \vec{R}_{\beta}|} - \sum_{i, \alpha} \frac{Z_{\alpha} e^2}{|\vec{R}_{\alpha} - \vec{r}_i|} + \frac{1}{2} \sum_{i, j} \frac{e^2}{|\vec{r}_i - \vec{r}_j|}$$

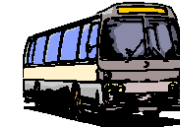
Kinetic energy of nuclei Kinetic energy of electrons Nucleus-Nucleus interaction Electron-Nucleus interaction Electron-Electron interaction

$$H\Psi = E\Psi$$

Ab-initio (first principles) Method – ONLY Atomic Numbers $\{Z_i\}$ as input parameters

Materials Science: Why *ab-initio* approach is needed?

- A model as simple as possible
- A model non-empirical and realistic



- Explanation and extrapolation of experimental results
- Physical insight
- Qualitative physics
- Reliable predictions of matter's properties
- Design of new materials
- Unexpected phenomena and unusual conditions (e.g., extreme pressures)

Quantum Mechanics of Molecules and Crystals

Molecule or Crystal = a system of **nuclei (ions)** and **electrons**

Nuclei – mass M , coordinates X , and momenta P , $X \equiv \{\vec{R}_1, \vec{R}_2, \dots, \vec{R}_{N_{nuc}}\}$
 (M, X, P)

Electrons – (m, x, p) $x \equiv \{\vec{r}_1, \vec{r}_2, \dots, \vec{r}_N\}$

$$H = \hat{T}_{el} + U(x, X) + \hat{T}_{Nuc}$$

Kinetic energy of electrons

$$\hat{T}_{el} = \sum_{i=1}^N \frac{1}{2m} \vec{p}_i^2 = -\sum_{i=1}^N \frac{\hbar^2}{2m} \nabla_i^2$$

Kinetic energy of the nuclei

$$\hat{T}_{Nuc} = \sum_{\alpha=1}^{N_{nuc}} \frac{1}{2M} \vec{P}_{\alpha}^2 = -\sum_{\alpha=1}^{N_{nuc}} \frac{\hbar^2}{2M_{\alpha}} \nabla_{\alpha}^2$$

Potential energy = The total Coulomb energy of nuclei and electrons

$$U(x, X) = \hat{V}_{en}(x, X) + \hat{V}_{ee}(x) + \hat{V}_{NN}(X)$$

Electron-nucleus

$$\hat{V}_{en}(x, X) = \sum_{i\alpha} \frac{-Z_{\alpha} e^2}{|\vec{r}_i - \vec{R}_{\alpha}|}$$

Electron-Electron

$$\hat{V}_{ee}(x) = \sum_{i < j} \frac{e^2}{|\vec{r}_i - \vec{r}_j|}$$

Nucleus-Nucleus

$$\hat{V}_{NN}(X) = \sum_{a < b} \frac{e^2}{|\vec{R}_a - \vec{R}_b|}$$

The Adiabatic Approximation (Born-Oppenheimer)

M. Born & J. R. Oppenheimer, Ann. Phys. 84, 457 (1927)

- It is natural to consider the full Hamiltonian of the system to be the sum of an **ionic** and an **electronic** part

$$\hat{H} = \hat{H}_N + \hat{H}_{el}$$

$$\hat{H}_N = \hat{T}_{Nuc} + \hat{V}_{NN}(X)$$

$$\hat{H}_{el} = \hat{T}_{el} + \hat{V}_{en}(x, X) + \hat{V}_{ee}(x)$$

The Adiabatic Approximation (Born-Oppenheimer)

- The Schrödinger equation for the electrons in the presence of fixed ions

$$\hat{H}_e \Psi_n(X, x) = E_n(X) \Psi_n(X, x)$$

Parametric dependence on ionic positions

- The energy levels of the system of ions are determined by solving

$$[\hat{H}_N + E(K', X)] \chi(Q, K', X) = \varepsilon(Q) \chi(Q, K', X)$$

The electronic energy contributes to the potential energy of the ion system. This implies that the potential energy depends on the state of the electrons.

Quantum Mechanics: System of N electrons in an external potential

- Adiabatic approximation – interacting electrons move in the 'external' potential of nuclei (ions) at fixed positions

$$\hat{T} = \sum_{i=1}^N -\frac{\hbar^2}{2m} \nabla_i^2 \quad \hat{H} = \hat{T} + \hat{V}_{en} + \hat{V}_{e-e} \quad \hat{V}_{e-e} = \sum_{i < j} \frac{e^2}{|\vec{r}_i - \vec{r}_j|} \quad \{ \vec{R}_1, \vec{R}_2, \dots \}$$

$$\hat{V}_{en} = \sum_{ia} \frac{-Z_a e^2}{|\vec{r}_i - \vec{R}_a|} = \hat{V}_{ext} = \sum_i v_{ext}(\vec{r}_i) \quad \left(E_{nn} = \sum_{a < b} \frac{Z_a Z_b e^2}{|\vec{R}_a - \vec{R}_b|} \right)$$

Schrödinger equation $\Phi(\{ \vec{R}_\alpha \}, \vec{r}_1, \vec{r}_2, \dots, \vec{r}_N) \equiv \Phi(\vec{r}_1, \vec{r}_2, \dots, \vec{r}_N)$

$H\Psi = E\Psi$ Many particle wave function $N \approx 10^{23}$

Ritz Variational Principle → Ground State Energy of the system

$$E_0 = \min_{\Psi \rightarrow N} \langle \Psi | \hat{H} | \Psi \rangle = \min_{\Psi \rightarrow N} \langle \Psi | \hat{T} + \hat{V}_{e-e} + \hat{V}_{ext} | \Psi \rangle \quad E[\Psi] = \frac{\langle \Psi | \hat{H} | \Psi \rangle}{\langle \Psi | \Psi \rangle}$$

$$\Psi(\vec{r}_1, \vec{r}_2, \dots, \vec{r}_N) \quad \text{Many-particle wavefunction} \quad E[\Psi] \geq E_0$$

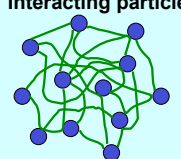
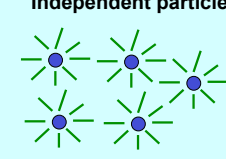
Full minimization of the functional $E[\Psi]$ with respect to all allowed N-electron wave functions

Quantum Mechanics: System of N electrons in an external potential

Schrödinger equation
 $H\Psi = E\Psi$

Exact analytical solutions are not known even for two electrons !

- Approximations are needed !
- Concept of independent particles moving in an effective potential

Interacting particles	Independent particles
	
$\Psi(\vec{x}_1, \vec{x}_2, \dots, \vec{x}_{10^{23}})$	$\varphi(\vec{x}_1) \varphi(\vec{x}_2) \dots \varphi(\vec{x}_{10^{23}})$

Idea: consider electrons as independent particles moving in an effective potential

Hartree and Hartree-Fock Approximation

Ansatz for the wave-function

Hartree Method

$$\Phi_{Hartree}(\vec{x}_1, \vec{x}_2, \dots, \vec{x}_N) = \varphi_1(\vec{x}_1) \varphi_2(\vec{x}_2) \dots \varphi_N(\vec{x}_N)$$

Hartree-Fock Method

$$\Phi_{H-F}(\vec{x}_1, \vec{x}_2, \dots, \vec{x}_N) = \frac{1}{\sqrt{N!}} \begin{vmatrix} \varphi_1(\vec{x}_1) & \varphi_2(\vec{x}_1) & \dots & \varphi_N(\vec{x}_1) \\ \varphi_1(\vec{x}_2) & \varphi_2(\vec{x}_2) & \dots & \varphi_N(\vec{x}_2) \\ \vdots & \vdots & \ddots & \vdots \\ \varphi_1(\vec{x}_N) & \varphi_2(\vec{x}_N) & \dots & \varphi_N(\vec{x}_N) \end{vmatrix}$$

ψ_i - one-electron wavefunction of the i th level

Hartree-Fock Approximation

$$\Phi_{H-F} \rightarrow E[\Phi_{H-F}] = \frac{\langle \Phi_{H-F} | \hat{H} | \Phi_{H-F} \rangle}{\langle \Phi_{H-F} | \Phi_{H-F} \rangle}$$

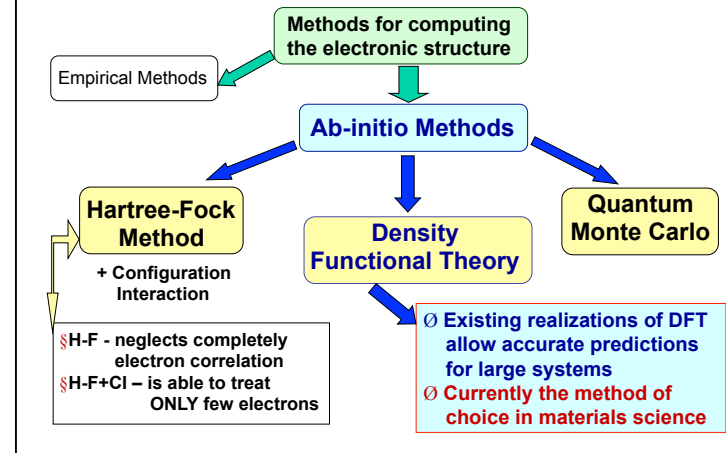
$$H = H_0 + \frac{1}{2} \sum_{i,j} U(\vec{x}_i, \vec{x}_j)$$

$$H_0 = \sum_i H_0(i) = \sum_i -\frac{1}{2} \nabla_i^2 + V_{ext}(\vec{r}_i) \quad U(\vec{x}_i, \vec{x}_j) = \frac{1}{|\vec{r}_i - \vec{r}_j|}$$

Variational Principle

$$\rightarrow H_0 \phi_i(\vec{x}_i) + \left[\sum_{j=1}^N \int \phi_j^*(\vec{x}_j) U(\vec{x}_i, \vec{x}_j) \phi_j(\vec{x}_j) d\vec{x}_j \right] \phi_i(\vec{x}_i) - \left[\sum_{j=1}^N \int \phi_j^*(\vec{x}_j) U(\vec{x}_i, \vec{x}_j) \phi_i(\vec{x}_j) d\vec{x}_j \right] \phi_j(\vec{x}_i) = \epsilon_i \phi_i(\vec{x}_i)$$

Spectrum of Electronic Hamiltonian: What *ab initio* methods do we have?



Density Functional Theory (DFT)

P. Hohenberg and W. Kohn, Phys. Rev. 136, B864 (1964)

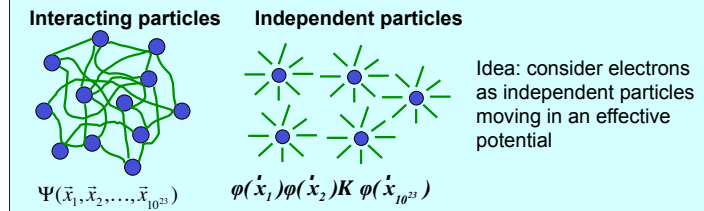
- One particle density – Basic quantity of DFT

$$\rho(\vec{r}) = \left\langle \Psi(\vec{r}_1, \vec{r}_2, \dots, \vec{r}_N) \left| \sum_i \delta(\vec{r}_i - \vec{r}) \right| \Psi(\vec{r}_1, \vec{r}_2, \dots, \vec{r}_N) \right\rangle = N \int d\vec{r}_2, \dots, d\vec{r}_N \Psi^*(\vec{r}, \vec{r}_2, \dots, \vec{r}_N) \Psi(\vec{r}, \vec{r}_2, \dots, \vec{r}_N)$$

- The DFT is based on two fundamental theorems for a functional of the one particle density.
- One particle density determines the ground state energy of the system
- Modern formulation – constrained-search method of Mel Levy

Mel Levy, Proc. Natl. Acad. Sci. USA, vol. 76, No. 12, p.606 (1979).

Density Functional Theory (DFT) in Kohn-Sham realization



This reduction is rigorously possible !

Density Functional Theory – constrained search formulation

Mei Levy, Proc. Natl. Acad. Sci. USA, vol. 76, No. 12, p.606 (1979).

Functional of the one particle density $F[\rho] \doteq \min_{\Psi \rightarrow \rho} \langle \Psi | \hat{T} + \hat{V}_{e-e} | \Psi \rangle$

The functional $F[\rho]$ searches all many particle functions Ψ that yield the input density $\rho(\vec{r})$ and then delivers the minimum of $\langle \hat{T} + \hat{V}_{e-e} \rangle$

Theorem I $\int d\vec{r} v_{ext}(\vec{r})\rho(\vec{r}) + F[\rho] \geq E_0$

Theorem II $\int d\vec{r} v_{ext}(\vec{r})\rho_0(\vec{r}) + F[\rho_0] = E_0$ ρ_0 - ground state density
 E_0 - ground state energy

Let us define function Ψ_{min}^ρ that minimizes $\langle \Psi | \hat{T} + \hat{V}_{e-e} | \Psi \rangle$

$F[\rho] = \langle \Psi_{min}^\rho | \hat{T} + \hat{V}_{e-e} | \Psi_{min}^\rho \rangle$ $F[\rho_0] = \langle \Psi_{min}^{\rho_0} | \hat{T} + \hat{V}_{e-e} | \Psi_{min}^{\rho_0} \rangle$

Proof of Theorem I:

$\int d\vec{r} v_{ext}(\vec{r})\rho(\vec{r}) + F[\rho] = \int d\vec{r} v_{ext}(\vec{r})\rho(\vec{r}) + \langle \Psi_{min}^\rho | \hat{T} + \hat{V}_{e-e} | \Psi_{min}^\rho \rangle = \langle \Psi_{min}^\rho | \hat{V}_{ext} + \hat{T} + \hat{V}_{e-e} | \Psi_{min}^\rho \rangle \geq E_0$

Ritz variational principle

Density Functional Theory – constrained search formulation

Proof of Theorem II:

$E_0 \leq \langle \Psi_{min}^{\rho_0} | \hat{V}_{ext} + \hat{T} + \hat{V}_{e-e} | \Psi_{min}^{\rho_0} \rangle$

From variational principle

$\langle \Psi_0 | \hat{V}_{ext} + \hat{T} + \hat{V}_{e-e} | \Psi_0 \rangle \leq \langle \Psi_{min}^{\rho_0} | \hat{V}_{ext} + \hat{T} + \hat{V}_{e-e} | \Psi_{min}^{\rho_0} \rangle$
 $\int d\vec{r} v_{ext}(\vec{r})\rho_0(\vec{r}) + \langle \Psi_0 | \hat{T} + \hat{V}_{e-e} | \Psi_0 \rangle \leq \int d\vec{r} v_{ext}(\vec{r})\rho_0(\vec{r}) + \langle \Psi_{min}^{\rho_0} | \hat{T} + \hat{V}_{e-e} | \Psi_{min}^{\rho_0} \rangle$
(A) $\langle \Psi_0 | \hat{T} + \hat{V}_{e-e} | \Psi_0 \rangle \leq \langle \Psi_{min}^{\rho_0} | \hat{T} + \hat{V}_{e-e} | \Psi_{min}^{\rho_0} \rangle$

But, on the other hand, from the definition of $\Psi_{min}^{\rho_0}$

(B) $\langle \Psi_0 | \hat{T} + \hat{V}_{e-e} | \Psi_0 \rangle \geq \langle \Psi_{min}^{\rho_0} | \hat{T} + \hat{V}_{e-e} | \Psi_{min}^{\rho_0} \rangle$

[(A) & (B) true] $\Rightarrow \langle \Psi_0 | \hat{T} + \hat{V}_{e-e} | \Psi_0 \rangle = \langle \Psi_{min}^{\rho_0} | \hat{T} + \hat{V}_{e-e} | \Psi_{min}^{\rho_0} \rangle$

$F[\rho_0] = \langle \Psi_{min}^{\rho_0} | \hat{T} + \hat{V}_{e-e} | \Psi_{min}^{\rho_0} \rangle$

$\int d\vec{r} v_{ext}(\vec{r})\rho_0(\vec{r}) + \langle \Psi_0 | \hat{T} + \hat{V}_{e-e} | \Psi_0 \rangle = F[\rho_0] + \int d\vec{r} v_{ext}(\vec{r})\rho_0(\vec{r})$

$\langle \Psi_0 | \hat{V}_{ext} + \hat{T} + \hat{V}_{e-e} | \Psi_0 \rangle = F[\rho_0] + \int d\vec{r} v_{ext}(\vec{r})\rho_0(\vec{r})$

$E_0 = F[\rho_0] + \int d\vec{r} v_{ext}(\vec{r})\rho_0(\vec{r})$

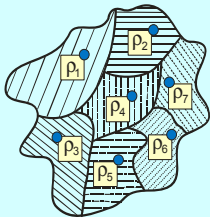
Density Functional Theory – Constrained Search Formulation Relation to Ritz Variational Principle

The ground-state energy minimization procedure of $E[\Psi] = \frac{\langle \Psi | \hat{H} | \Psi \rangle}{\langle \Psi | \Psi \rangle}$ can be divided into two steps

$E_0[\Psi] = \min_{\Psi \rightarrow N} \langle \Psi | \hat{T} + \hat{V}_{e-e} + \hat{V}_{ext} | \Psi \rangle = \min_{\rho \rightarrow N} \left[\min_{\Psi \rightarrow \rho} \langle \Psi | \hat{T} + \hat{V}_{e-e} + \hat{V}_{ext} | \Psi \rangle \right]$

- The inner minimization is constrained to all wave functions that give $\rho(\vec{r})$,
- while the outer minimization releases this constrain by searching all $\rho(\vec{r})$

Percus-Levy partition of the N-electron Hilbert space



- Each shaded area is the set of Ψ that integrate to a particular $\rho(\vec{r})$.
- The minimization $\Psi \rightarrow \rho$ for a particular ρ is constrained to the shaded area associated with this ρ , and is realized by one point (denoted by \bullet) in this shaded area.
- The minimization $\rho \rightarrow N$ is over all such points.

Density Functional Theory – Constrained Search Formulation Relation to Ritz Variational Principle

■ $E_0[\Psi] = \min_{\Psi \rightarrow N} \langle \Psi | \hat{T} + \hat{V}_{e-e} + \hat{V}_{ext} | \Psi \rangle = \min_{\rho \rightarrow N} \left[\min_{\Psi \rightarrow \rho} \langle \Psi | \hat{T} + \hat{V}_{e-e} + \hat{V}_{ext} | \Psi \rangle \right] = \min_{\rho \rightarrow N} \left[\min_{\Psi \rightarrow \rho} \langle \Psi | \hat{T} + \hat{V}_{e-e} | \Psi \rangle + \int d\vec{r} v_{ext}(\vec{r})\rho(\vec{r}) \right] = \min_{\rho \rightarrow N} [F[\rho] + \int d\vec{r} v_{ext}(\vec{r})\rho(\vec{r})] = \min_{\rho \rightarrow N} E[\rho]$ $E[\rho] = F[\rho] + \int d\vec{r} v_{ext}(\vec{r})\rho(\vec{r})$

In ■ 2^N wave functions of 3N variables ☹️

In ● ONE function of 3 variables !!! 😊

Density Functional Theory

PROBLEM: exact functional $F[\rho]$ is unknown!
One needs a good approximation to $F[\rho]$

$$F[\rho] = \min_{\Psi_\rho} \langle \Psi_\rho | \hat{T} + \hat{V}_{e-e} | \Psi_\rho \rangle = \langle \Psi_{min}^\rho | \hat{T} + \hat{V}_{e-e} | \Psi_{min}^\rho \rangle$$

$$= T[\rho] + U[\rho] + \left\{ \langle \Psi_{min}^\rho | \hat{V}_{e-e} | \Psi_{min}^\rho \rangle - U[\rho] \right\}$$

Kinetic energy

Exchange & Correlation $E_{xc}[\rho]$

Classical Coulomb energy

$$U[\rho] = \frac{1}{2} \iint d\vec{r} d\vec{r}' \frac{\rho(\vec{r})\rho(\vec{r}')}{|\vec{r} - \vec{r}'|}$$

$$F[\rho] = T[\rho] + U[\rho] + E_{xc}[\rho]$$

- The functional $F[\rho]$ is universal in the sense that it is independent of the external potential (field) $v_{ext}(\vec{r})$.

Thomas-Fermi-Method (probably the oldest approximation to DFT)

$$T^{T-F}[\rho] \cong \frac{3}{5} (3\pi^2)^{2/3} \frac{\hbar^2}{2m} \int d\vec{r} |\rho(\vec{r})|^{5/3} \quad V_{ee}^{T-F}[\rho] \equiv U[\rho]$$

and extensions

- § Thomas-Fermi-Dirac
- § Thomas-Fermi-Weizsacker

PROBLEM: $T^{T-F}[\rho]$

Very often these models give even qualitatively wrong results.

DFT- The Kohn- Sham Method

W. Kohn and L. J. Sham, *Phys. Rev.* **140**, A1133 (1965)

W. Kohn & L. Sham (1965) invented an ingenious indirect approach to the kinetic- energy functional.

They turned density functional theory into a practical tool for rigorous calculations

The main idea:

System of interacting electrons with density $\rho(\vec{r})$

System of non-interacting electrons with the same density $\rho(\vec{r})$

"Real" system

$\rho(\vec{r})$ $T[\rho]$

"Fictitious" or Kohn-Sham reference system

$T_s[\rho]$ $\rho_s(\vec{r}) = \rho(\vec{r})$

$$E[\rho] = \int d\vec{r} v_{ext}(\vec{r})\rho(\vec{r}) + T_s[\rho] + U[\rho] + E_{xc}[\rho]$$

- $E_{xc}[\rho] = V_{ee}[\rho] - U[\rho] + T[\rho] - T_s[\rho]$
- Exchange-correlation functional** contains now the difference between kinetic energy functional of interacting and non-interacting electrons.

The Kohn- Sham Method – Kinetic energy functional

How the $T_s[\rho]$ looks like ?

Hamiltonian of the non-interacting reference system

$$H_s = \sum_i \left[-\frac{\hbar^2}{2m} \vec{\nabla}_i^2 + \sum_s v_s(\vec{r}_i) \right] \quad v_s(\vec{r}) \text{ - local potential}$$

For this system there will be an

exact determinantal ground-state wave function

$$\Phi = \frac{1}{\sqrt{N!}} \det[\phi_1, \phi_2, \dots, \phi_N] \quad \text{where } \phi_i \text{ are the } N \text{ lowest eigenstates of the one-electron Hamiltonian}$$

$$\hat{h}_s \phi_i = \left[-\frac{\hbar^2}{2m} \vec{\nabla}^2 + v_s(\vec{r}) \right] \phi_i(\vec{r}) = \epsilon_i \phi_i(\vec{r}) \quad \text{The density}$$

$$\rho(\vec{r}) = \sum_{i=1}^N \phi_i^*(\vec{r}) \phi_i(\vec{r})$$

The Kohn- Sham Method – Kinetic energy functional

$T_s[\rho]$ - can be defined by the constrained-search formula

$$T_s[\rho] = \text{Min}_{\Phi \rightarrow \rho} \langle \Phi | \hat{T} | \Phi \rangle = \text{Min}_{\Phi \rightarrow \rho} \sum_{i=1}^N \left\langle \phi_i \left| -\frac{\hbar^2}{2m} \vec{\nabla}^2 \right| \phi_i \right\rangle$$

The search is over all single-determinantal functions Φ that yield the given density ρ .

- The existence of the minimum has been proved by Lieb (1982).
- $T_s[\rho]$ is uniquely defined for any density.
- $T_s[\rho] \neq T[\rho]$

Crucial characteristics of the Kohn-Sham Method

Φ NOT Ψ

Thank you!

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